





## CORROSION DAMAGE AND DEGRADATION

**N. Mary** (09.2023 – Tohoku Univ.)





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## 1. PART 1. INTRODUCTION OF DAMAGE AND DEGRADATION



#### 2. CORROSION BASIS

3. CORROSION TESTING, MONITORING, INSPECTION

4. CORROSION PREVENTION



#### INTRODUCTION

Most commonly used metals are unstable in the atmosphere (except gold and platinium)

Corrosion is the process of a metal returning to the material's thermodynamic state.

For most materials its means the formation of oxide or sulfides

These changes are electrochemical reactions that follow the laws of thermodynamics.

 $\rightarrow$  added dimension of chemistry and electricity

 $\rightarrow$  reaction time and temperature dependent

 $\rightarrow$  corrosion rate affected by ion and corrodent concentration

→this explain why some reaction are reversible or controllabe while other not

#### INTRODUCTION

Two parameters of the system (material) change:

- Gibbs free Energy  $\Delta G$ :
  - Thermodynamical or chemical potential which reduces when a system attains equilibrium.
  - Major factor on the spontaneity of reduction-oxidation reaction in electrochemistry

More negative of  $\Delta G$ , the greater the tendency for the reaction to go

$$Mg + H_2O + \frac{1}{2}O_2 \to Mg(OH)_2 \qquad \Delta G^0_{25^\circ C} = -596 \ kJ$$
$$Cu + H_2O + \frac{1}{2}O_2 \to Cu(OH)_2 \qquad \Delta G^0_{25^\circ C} = -119 \ kJ$$
$$Au + \frac{3}{2}H_2O + \frac{3}{4}O_2 \to Au(OH)_3 \qquad \Delta G^0_{25^\circ C} = +65 \ kJ$$

Note that large negative  $\Delta G$  may or may not be rapid  $\rightarrow$  kinetic !

#### INTRODUCTION

Two parameters of the system (material) change:

- Gibbs free Energy  $\Delta G$ :
- Pilling-Bedworth ratio (note it exists exceptions and limitations)
  - Parameter (1923) derived from metal oxidation in High Temperature
  - Metal-oxide PB ratio describe as the ratio of the metal oxide volume produced by the oxygen and the metal reaction of the consumed value of metal

$$R_{PB} = \frac{V_{oxide}}{V_{metal}} = \frac{M_{oxide} \times \rho_{metal}}{n \times M_{metal} \times \rho_{oxide}}$$

- If  $R_{PB} < 1$ : oxide film too thin  $\rightarrow$  break down (Mg case)
- If  $1 < R_{PB} < 2$ : oxide film passivating  $\rightarrow$  passive films (Ti,Cr, Al,...)
- If  $R_{PB} > 2$ : oxide film chips off  $\rightarrow$  no protection (Fe, Cu)

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Metal vs environment: High temperature

- At low temperature : diffusion of  $O_2$  and metal through compact oxide film
- At moderate and high temperature:
  - combination of oxide film formation and oxide volatility
  - Formation of volatile metal and oxide species at M-O interface, transport through oxide lattice and mechanically formed cracks in oxide layer
  - Direct formation of volatile oxide gases
- At high temperature
  - Gaseous diffusion of oxygen through a barrier layer of volatilized oxides
  - Spalling of metal and oxide particles



Figure III - 2 : Mécanisme de l'oxydation d'un métal à haute température

Linear law (Ba, Ma)  $\xi = kt$ Parabolic law (Fe, Ni, Cu)  $\xi^2 = 2kt$ Log law (Zn)  $\xi = \xi_0 + k \ln(t)$ 



Figure 1.36 Oxidation rate loss showing weight gain and weight loss with different kinetic forms. (Reproduced from Corrosion for Science and Engineering, Tretheway and Chamberlain, Copyright Pearson Education Ltd)

Corrosion Prevention and Protection, Practical Solutions. V.S. Sastri (2007)

Metal vs environment: aqueous environment

- Water adsorption at the metal surface

oxidation

Atmospheric  $(H_20, SO_2, Cl^-)$ 





oxidation

Figure 4.3 Effect of  $SO_2$  and humidity on metallic corrosion. Reaction occur in a very thin (invisible) aqueous layer

Matériaux métalliques phénomènes de corrosion. M. Hélie. Cefracor Principle of Corrosion Engineering and corrosion control. Z. Ahmad. (2006) Corrosion Engineering, Principles and Pratice. P.R. Robergue (2008)







FIGURE 9.37 Corrosivity map of Japan.

Metal vs environment: aqueous environment

- Water adsorption at the metal surface
  - Aqueous solution





+ + +

Metal vs environment: <u>aqueous / atmospheric environment</u>



Figure IV - 9 : Exemples de corrosion par aération différentielle

Matériaux métalliques phénomènes de corrosion. M. Hélie. Cefracor http://www.corrosionengineering.co.uk/knowledge-library/aqueous-corrosion/index.php

#### **AQUEOUS CORROSION**

Most basic corrosion reactions involves the oxidation of a pure metal when exposed to a strong acid.

For exemple Iron exposed to hydrochloric acid :

 $Fe + 2 HCl \rightarrow FeCl_2 + H_2$ 

The result of this reaction is the gradual disappears of the iron and the dihydrogen bubbles rising rapidly.

On an electrochemical level, there is also an exchange of electron  $Fe + 2H^+ + Cl^{2-} \rightarrow Fe^{2+} + Cl^{2-} + H_2$ 

- Iron is converted to an iron ion:  $Fe \rightarrow Fe^{2+} + 2e$
- Hydrogen gain electrons:  $2H^+ + 2e \rightarrow H_2$

#### **ANODE AND CATHODE**

Within electrode, current is carried by both negative and positive carriers

- Negative carriers are electrons and move in a metal
- Positive carriers are ions in solution and current carried by ions depends on its mobility and electric charge of the ions

The reservoir of negative is the metal and release electron by an oxidation reaction

$$Zn \rightarrow Zn^{2+} + 2e$$
  $do(Zn) = 0 \rightarrow do(Zn^{2+}) = +II$ 

The positive carrier are ions, molecules or metal able to attract electrons of the oxidation

$$2H^+ + 2e \rightarrow H_2$$
$$Cu^{2+} + 2e \rightarrow Cu$$



 $Zn \rightarrow Zn^{2+} + 2e$  (anode: oxidation)  $2H^+ + 2e \rightarrow H_2$  (cathode: reduction)

#### **CATHODIC PROCESSES**

Reduction of hydrogen ions at the cathodic surface disturb the balance between the acidic ions and hydroxyl ions and make the solution less acidic or more alkaline at the corrosion surface

In neutral solution, [AI, Zn, Mg] develops enough energy to split water directly as  $2H_2O (liq) + 2 e \rightarrow H_2 + OH^-$ 

Several other cathodic reactions occurs

Dissolved dioxygen reduction (acid) $O_2 + 4H^+ + 4e \rightarrow 2H_2O$ Dissolved dioxygen reduction (neutral) $O_2 + 2H_2O + 4e \rightarrow 4OH^-$ Metal reduction $M^{n+} + n e \rightarrow M$ Hydrogen sulfate reductions $2H_2S + 2e \rightarrow H_2 + 2HS^-$ Nitric acid reduction $HNO_3 + 3H^+ + 3e \rightarrow NO + 2H_2O$ etc.

#### **CELL POTENTIAL**

A reaction will occur only is there is a negative free energy change ( $\Delta G$ )

 $\Delta G = -n FE$ 

Where n is the number of electrons, F is the Faraday constant and E is the cell potential

 $Fe \rightarrow Fe^{2+} + 2e$  $2H^+ + 2e \rightarrow H_2$ Oxidation (loss of electron)reduction (gain of electrons)

Each reaction has his own cell potential  $E = E(H^+/H_2) - E(Fe^{2+}/Fe) = 0 - (-0.44) = +0.44$ 

Because the cell potential is positive, the reaction takes place



#### <u>Figure 3.3.</u> Galvanic series in seawater [5]. (Reprinted with permission of ASM International<sup>®</sup>. All rights reserved. www.asminternational.org.)

#### **CELL POTENTIAL**

#### TABLE 2.1

Standard Oxidation–Reduction Potentials 25°C, Volts vs. Hydrogen Electrode

Redox Reaction	Oxidation- Reduction Potential
$Au = Au^{3+} + 3e^{-}$	+1.498
$O_2 + 4H^+ + 4e^- = 2H_2O$	+1.229
$Pt = Pt^{2+} + 2e^{-}$	+1.2
$Pd = Pd^{2+} + 2e^{-}$	+0.987
$Ag = Ag^+ = e^-$	+0.799
$2Hg = Hg_2^{2+} + 2e^{-1}$	+0.788
$Fe^{3+} + e^- = Fe^{2+}$	+0.771
$O_2 + 2H_2O + 4e^- = 4OH^-$	+0.401
$Cu = Cu^{2+} + 2e^{-}$	+0.337
$Sn^{4+} + 2e^- = Sn^{2+}$	+0.15
$2H^+ + 2e^- = H_2$	0.0000
$Pb = Pb^{2+} + 2e^{-}$	-0.126
$Sn = Sn^{2+} + 2e^{-}$	-0.136
$Ni = Ni^{2+} + 2e^{-}$	-0.250
$Co = Co^{2+} + 2e^{-}$	-0.277
$Cd = Cd^{2+} + 2e^{-}$	-0.403
$Fe = Fe^{2+} + 2e^{-}$	-0.440
$Cr = Cr^{3+} + 3e^{-}$	-0.744
$Zn = Zn^{2+} = 2e^{-}$	-0.763
$Al = Al^{3+} + 3e^{-}$	-1.662
$Mg = Mg^{2+} + 2e^{-}$	-2.363
$Na = Na^+ + e^-$	-2.714
$\mathbf{K} = \mathbf{K}^+ + \mathbf{e}^-$	-2.925

Corrosion and corrosion control. R.W. Revie, H.H. Uhlig, (2007)

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How behave a brass in either  $H_2SO_4$  or  $HNO_3$  solution?

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1- Environment

both solutions are acid.  $H^+$  has to be considered rather than  $O_2$ 

 $SO_4^{2-}$  is rather stable in solution whereas  $NO_3^-$  is in equilibrium with *NO* according  $NO_3^- + 4 H^+ + 3 e \rightarrow NO + 2 H_2O$ 

 $\rightarrow$  different behavior are expected because of nature of chemical

+ + +

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2- Material

2 elements can oxidize  $Cu \rightarrow Cu^{2+} + 2e$  and  $Zn \rightarrow Zn^{2+} + 2e$ 

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3- Potential scale

 $NO_{3}^{-}/NO > Cu^{2+}/Cu > H^{+}/H_{2} > Zn^{2+}/Zn$ 



+ + +

+ + +

+ + +





Only 2 reactions take place  $2H^+ + 2e \rightarrow H_2$  $Zn \rightarrow Zn^{2+} + 2e$ 



+ + +

+ + +

+ + +

 $HNO_3$ 

 $NO_3^{-}$  NO *Cu*<sup>2+</sup> Cu  $H^+$  $H_2$  $Zn^{2+}$  Zn

+ + +



Several equations takes places  $2H^+ + 2e \rightarrow H_2$  $Zn \rightarrow Zn^{2+} + 2e$  $NO_3^- + 4 H^+ + 3 e \rightarrow NO + 2 H_2O$  $Zn \rightarrow Zn^{2+} + 2e$  $NO_3^- + 4H^+ + 3e \rightarrow NO + 2H_2O$  $Cu \rightarrow Cu^{2+} + 2e$ 

Zinc + copper electro-dissolution

#### Consequences on the material surface



### **TYPE OF LOCAL CELL FORMATION**

Three main types of cells in corrosion reactions

• Dissimilar Electrodes Cells

Including different materials, similar materials but different mechanical states (cold work metal vs. same annealed metal, grain boundary vs. grains, different crystal orientation.



#### **TYPE OF LOCAL CELL FORMATION**

Three main types of cells in corrosion reactions

• Concentration cells

Two identical electrodes in contact with different solution.

Salt concentration cell: same solution but different concentration metal ion concentration

differential aeration cell : dissolve oxygen concentration vs deareated



Figure 2.3. Salt concentration cell.



Figure 2.5. Differential aeration cell formed by rust on iron.



Figure 2.6. Water-line corrosion, showing differential aeration cell.

Corrosion and corrosion control. R.W. Revie, H.H. Uhlig, (2007)

## **TYPE OF LOCAL CELL FORMATION**

Three main types of cells in corrosion reactions

• Differential temperature cells

Found in heat exchangers, boilers, immersion heaters...

Same material but different temperature





Figure 7.14 Thermogalvanic corrosion.

Corrosion and Protection. E. Bardal (2004) https://corrosion-doctors.org/Corrosion-Factors-Cells/corrosion-cells-temperature.htm

Potential-pH diagrams or Pourbaix diagrams

Graphical représentations of the thermodynamical stability of a chemical specie

Application to water stability

Two couples  $H_2 O/H_2$  and  $O_2/H_2 O$  which can be written as:

And potentiel are written according to the Nernst equation

$$\Phi = \Phi^{\circ} - 2.303 \frac{RT}{1F} \log \frac{[\mathrm{H}^{+}]}{\mathrm{P}_{\mathrm{H}_{2}}^{\frac{1}{2}}} = 0.06 \log(\mathrm{H}^{+}) = -0.06 \,\mathrm{pH}$$
$$\Phi = \Phi^{\circ} - 2.303 \frac{RT}{1F} \log \frac{[\mathrm{H}^{+}]^{2} \cdot (Po_{2})^{\wedge}(\frac{1}{2})}{1} = 1.23 - 0.06 \,\mathrm{pH}$$

$$\begin{split} H_2 O &\rightarrow \frac{1}{2} O_2 + 2 H^+ + 2 e \\ H^+ + e &\rightarrow \frac{1}{2} H_2 \\ H_2 O + e &\rightarrow \frac{1}{2} H_2 + O H^- \end{split}$$

Corrosion and Corrosion control. R Winston Revie, H. H. Uhlig. (4th edition 2008)

Potential-pH diagrams or Pourbaix diagrams

Graphical representations of the thermodynamical stability of a chemical specie

B line:  $H_2O \rightarrow \frac{1}{2}O_2 + 2H^+ + 2e$ A line:  $2H^+ + 2e \rightarrow H_2$ 



Potential (volts, S.H.E.)



Horizontal line does not involve pH :  $Fe^{3+} + 3e \rightleftharpoons Fe$ 

Vertical line involves  $H^+$  or  $OH^-$  but not electron:  $Fe^{3+} + 3OH^- \rightarrow Fe(OH)_3$ 

oblique lines are driven by the Nernst equation:  $\Phi = \Phi^{\circ} - 2.303 \frac{RT}{nF} \log \left(\frac{[\text{Fe}^{3+}]}{[\text{Fe}]}\right)$ 

Iron Pourbaix diagram as a function of temperature



10.1016/S0010-938X(96)00067-4

Fig. 2. Pourbaix diagrams for iron at  $10^{-6}$  m at 25, 100, 200 and 300°C.



Matériaux métalliques phénomènes de corrosion. M. Hélie. Cefracor

#### **POTENTIAL PH DIAGRAMS**





Unstability domain of Au?



Mg Pourbaix diagram

Stability domain of magnesium?

Data indicate the conditions for which oxidehydroxyde are form but do not inform about their barrier effectiveness



Figure 1.5 E-pH diagram of iron in water at 25°C and its observed corrosion behavior.

## **CORROSION MONITORING – CORROSION RATE**

In practice, we are concerned with rates of corrosion (i.e. kinetic).

 Aluminium has a pronounced tendency to react however its rate is very slow so it meet the requirement of a structural metal.

Direct corrosion measurements techniques

Intrusive techniques

Physical techniques (mass loss, electrical resistance, visual,...)

Electrochemical techniques

Nonintrusive techniques

Physical techniques for metal loss (U.S., Eddy Current,...) Physical tech. for crack detection and propagation (A.E., Ultrasonic)

#### **CORROSION MONITORING – CORROSION RATE**



Electrochemical technique under estimate the mass loss



## **CORROSION MONITORING – CORROSION RATE**

indirect corrosion measurements techniques

On line techniques

hydrogen monitoring, corrosion potential water chemistry (pH, conductivity, etc.) fluide, Pressure, temperature Off line techniques water chemistry

residual inhibitor

chemical analysis of samples



The current releases by one anodic reaction can be convert an equivalent mass loss or corrosion penetration rate with the Faraday's law

 $Ag \rightarrow Ag^+ + e$ 

1 mol of silver oxidation leads to one mol of  $Ag^+$  and one mol of electron (i.e.  $6.022 \times 10^{23}$  electrons)

1 mol of electron is equal to 1 faraday (F) or 96,485 C/mol (of electron)

Combining Faraday's principles with specific electrochemical reaction of known stoichiometry leads to  $Q = F \times \Delta N \times n$  wheres  $\Delta N$  is the change in number of moles, *n* is the number of electrons per molecule of the species being reacted.

The charge Q can be defined in terms of electric current:

$$Q = \int_0^t I.\,dt$$

Where *I* is the total current in amperes (A), *t* the process duration in second

#### **CORROSION RATE**

The corrosion current itself can be either estimated by using electrochemical methods or by using weight loss data and a conversion chart based on Faraday principle.

	mA cm⁻²	mm y <sup>-1</sup>	тру	g m <sup>-2</sup> day <sup>-1</sup>
mA cm <sup>-2</sup>	1	3.28 M/nd	129 M/nd	8.95 <i>M/n</i>
mm y <sup>-1</sup>	0.306 nd/M	1	39.4	2.74 d
тру	0.00777 nd/M	0.0254	1	0.0694 <i>d</i>
g m <sup>-2</sup> day <sup>-1</sup>	0.112 n/M	0.365 /d	14.4 /d	1

where mpy = milli-inch per year

n = number of electrons freed by the corrosion reaction

M = atomic mass

d = density

\*Note: the table should be read from left to right, i.e.,

1 mA cm<sup>-2</sup> = (3.28 M/nd) mm y<sup>-1</sup> = (129 M/nd) mpy = (8.95 M/n) g m<sup>-2</sup> day<sup>-1</sup>

 TABLE 3.1
 Conversion between Current, Mass Loss, and Penetration Rates for all Metals\*

#### **CORROSION RATE**

#### Quelques grandeurs accessible

For uniform attack metals are classified into three groups according to their corrosion rates and intended application.

1. < 0.15 mm/y — Metals in this category have good corrosion resistance to the extent that they are suitable for critical parts valve seats, pump shafts and impellors, springs...

2. 0.15 to 1.5 mm/y— Metals in this group are satisfactory if a higher rate of corrosion can be tolerated, for example, for tanks, piping, valve bodies, and bolt heads.

3. > 1.5 mm/y ( > 0.05 ipy) — Usually not satisfactory.



#### **CORROSION UNDER OXYGEN REDUCTION**

Extensive and intensive parameters playing a rôle (as exemple)



Concentrations of oxygen in air-saturated water as a function of temperature [6.6]. (Reproduced from Tomashov ND. Theory of Corrosion and Protection of Metals. New York: Macmillan, 1966.)



Figure 6.4 Corrosion rate of steel in water as a function of temperature for an open system (a), a closed system (b), and a system where the oxygen concentration is kept constant (c).



Figure 6.8 Corrosion rate as a function of flow velocity. Steel in seawater at 23°C



Figure 6.11 Rate of atmospheric corrosion as a function of water film thickness (after Tomashov [6.6]).

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 $Figure 1.16\;$  Graphical representation of the processes occurring at an electrochemical interface.

$$D_o = \frac{RT\lambda_o}{|z_o|^2 F^2} \qquad (1.20)$$

where  $z_0$  = the valency of species O

 $R = {
m gas \ constant, \ i.e., \ 8.314 \ J \cdot mol^{-1} \cdot K^{-1}}$ 

T = absolute temperature, K

F = Faraday's constant, i.e., 96,487 C  $\cdot$  mol<sup>-1</sup>

Electrochemical techniques:

- Based on electron quantities, current, potential
- Direct information about corrosion rate or mechanism

Corrosion Engineering, Principles and Practice. P.R. Robergue (2008) Matériaux métalliques phénomènes de corrosion. M. Hélie. Cefracor

Electrochemical setup at the lab scale

3 electrodes cell :

- Anode electrode : metal in contact with the electrolyte on which anodic reaction takes places
- Cathode electrode : metal in contact with the electrolyte on which cathodic reaction takes places
- Potentiostat : providing the potential to the electrochemical system while counting the number of electron exchanged
- Electrolyte : electrically conductive fluid or moist soil



Figure 1.26 Electrochemical cell used for potentiodynamic polarization studies

#### Secondary reference electrodes





FIGURE 4.4

electrode.

Schematic of

a silver/silver

FIGURE 4.3 Graphical scheme to compare potentials of the most commonly used reference electrodes.

Potential measurements

(Rest potential, Open circuit potential, free potential)

Simplest measurement without electrical perturbation of the system

At this potential anodic and cathodic current are equals but different to zero

OCP evolution informs about the material evolution



Potential measurements

(Rest potential, Open circuit potential, free potential)

Simplest measurement without electrical perturbation of the system

At this potential anodic and cathodic current are equals but different to zero

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Fig. 2. OCP measurements on Ni100, Ni70, Cu100 and Ni30 in aerated synthetic groundwater. Temperature:  $25 \,^{\circ}$ C; pH: 8.8.

#### **ELECTROCHEMICAL TEST**

#### Current measurements : cathodic and anodic contributions







**FIGURE 2.10** Anodic and cathodic polarization curve.

TABLE 7.10	Examples of Maximum Scan Rates for Performing Valid Polarization
Plots	

Solution resistance, $\Omega \cdot cm^2$	$\begin{array}{c} Polarization \ resistance, \\ k\Omega \cdot cm^2 \end{array}$	Capacitance, μF·cm <sup>-2</sup>	$\begin{array}{c} \text{Maximum scan rate,} \\ m V {\cdot} s^{-1} \end{array}$
10	1	100	5.1
10	10	100	0.51
10	100	100	0.05
10	1000	100	0.005
100	1	100	6.3
100	10	100	0.51
100	100	100	0.05
100	1000	100	0.005
10	1	20	25
10	10	20	2.5
10	100	20	0.25
10	1000	20	0.025
100	1	20	50
100	10	20	2.6
100	100	20	0.25
100	1000	20	0.025

#### Fundamentals of corrosion. P.A. Schweitzer (2010) Handbook of corrosion engineering. P.R. Roberge (1999)

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The LPR is defined as  $\Delta E / \Delta i$  slope at the corrosion potential

With the Stern-Geary approximation  $R_p = \frac{B}{i_{corr}} = \left(\frac{\Delta E}{\Delta i}\right)_{\Delta E^{-1}}$ 

With 
$$B = \frac{b_a \cdot b_c}{2.3(b_a + b_c)}$$



FIGURE 5.21 Hypothetical linear polarization plot.

#### $b_a$ and $b_c$ are calculated from tafel slope or litterature

48

Current measurements : cathodic reaction (contration controlled processes)



Effet of mass transorport of  $O_2$  in weak acid solutions

49

Current measurements : anodic reaction modifiyed by the environment









Current measurements : anodic reaction modifiyed by the environment





 $\label{eq:Figure 5.4} Figure 5.4 \quad Anodic polarization curves for an 18-8 CrNi steel in a 0.1 M NaCl solution with different amounts of Na_2SO_4 added [5.3].$ 



Fundamentals of corrosion. P.A. Schweitzer (2010) Corrosion and Corrosion control. H.H. Uhlig (1971)

#### **ELECTROCHEMICAL TEST**

Current measurements : anodic reaction modified the surface reaction or itselft





Figure 5.2 Pourbaix diagram and anodic overvoltage curves for iron in water.

#### **ELECTROCHEMICAL TEST**

Current measurements : anodic reaction modified the surface reaction or itselft





I: Film is porous due to reduction dissolution of  $Fe_2O_3$ II: Film is compact without dissolution of both  $Cr_2O_3$  and  $Fe_2O_3$ III: Film is slightly porous due to field-assisted dissolution of  $Fe_2O_3$ IV: Film is porous due to oxidation dissolution of  $Cr_2O_3$ 

Fig. 14. Supposed structure of passive films on Fe-18Cr alloy in four potential ranges in 1 kmol  $m^{-3}$  H<sub>2</sub>SO<sub>4</sub>.

#### **ELECTROCHEMICAL TEST**

#### Current measurements : Electrochemical impendence measurements



#### Table 4

Impedance fitting parameters for samples exposed to aerated synthetic groundwater solution as a function of the nickel content in the alloy.

Parameters	Ni100	Ni70	Ni30	Cu100
EEC	$R_s - \begin{bmatrix} Q_d \\ R_t \end{bmatrix}$	$R_{s} - \begin{bmatrix} Q_{d} \\ R_{t} \end{bmatrix} $	$R_s - \begin{bmatrix} Q_f \\ R_f - \begin{bmatrix} \\ R_f \end{bmatrix}$	
$R_{\rm s}/\Omega~{\rm cm}^2$	55.3	55.2	55.2	60.8
$C_{ m dl}/\mu{ m F~cm^{-2}} Q_{ m dl}/{ m S~s}^n { m cm^{-2}} n_1 R_t/\Omega { m cm}^2 W/{ m S~s}^{1/2} { m cm}^{-2}$	21.5 $1.8 \times 10^{-5}$ 0.93 $8.9 \times 10^{5}$ -	$17.3 \\ 1.6 \times 10^{-5} \\ 0.91 \\ 1.1 \times 10^{5} \\ 5.3 \times 10^{-5}$	$72.2 \\ 1.8 \times 10^{-4} \\ 0.56 \\ 1.7 \times 10^{3}$	$\begin{array}{c} 199.9 \\ 9.1 \times 10^{-5} \\ 0.57 \\ 3.1 \times 10^{4} \end{array}$
$C_{\rm f}/\mu { m F}{ m cm}^2$ $Q_{\rm f}/{ m S}{ m s}^{ m n}{ m cm}^{-2}$ $n_2$ $R_{\rm f}/\Omega{ m cm}^2$	- - -	- -	$\begin{array}{c} 36.4 \\ 7.3 \times 10^{-5} \\ 0.87 \\ 1.0 \times 10^2 \end{array}$	$\begin{array}{c} 11.6 \\ 1.5 \times 10^{-5} \\ 0.92 \\ 1.9 \times 10^{3} \end{array}$
χ <sup>2</sup>	$1.52\times10^{-4}$	$8.03\times10^{-4}$	$6.16\times10^{-5}$	$\textbf{9.49}\times\textbf{10}^{-5}$



Fig. 5. (a) Nyquist and (b) phase impedance diagrams obtained for four alloy samples immersed in aerated synthetic groundwater. Temperature:  $25 \,^{\circ}$ C; pH 8.8, (•) exp, (–) fit.

10.3390/s21196578

Current measurements : Electrochemical impendence measurements



Fig. 3. The descriptive models used in the present methodology. Arabic numbers are the number of parameters used in the model.

- I: the oxide layer is taken into account without any hydroxide [19].
- II: the faradic reactions are limited by migration processes. The PDM of D. D. Macdonald demonstrates that the migration in passive film can be described by Warburg diffusion [20].
- III: the evolution of stoichiometry through the film leads to the evolution of conductivity in the oxide following an exponential law (Young model) [21], this non-faradic behavior is in parallel with a charge transfer resistance [22].
- IV: faradic reactions are limited by diffusion with flow equal to zero as boundary condition. It also corresponds to model VIII if the transport number of electrons is equal to the transport number of considered ions [17].
- V: faradic reactions are limited by diffusion in a finite diffusion layer. This model corresponds to model VIII if the transport of the positive and negative species have the same boundary conditions [23].
- VI: similar to I with a porous hydroxide layer [24];
- VII: similar to I with a non-porous hydroxide layer [25];
- VIII: faradic reactions are limited by diffusion / migration processes according to Vorotyntsev's calculation [26].
- IX: similar than V with a non-porous hydroxide layer;
- X: the MCM, developed by M. Bojinov, takes into account an outer interface with two charge transport paths through the oxide [27].



#### **TOWARD ELECTROCHEMICAL AT LOCAL SCALE**



https://doi.org/10.1021/acs.jpcc.2c04494



#### **SYNERGIES**

+ + +

**Mechanics** 

## **Materials**

Relations ?

@web

picture

@web picture



## Environment

@web picture